

# Polysulfone membranes for CO<sub>2</sub>/CH<sub>4</sub> separation: State of the art

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### ABSTRACT

The world market for  $CO_2/CH_4$  separation using membrane technology is getting bigger and polysulfone (PSf) is one of the most widely investigated polymer membrane materials. However, the existing PSf membranes performances are insufficient to fully exploit these opportunities. Therefore, modification that would produce membrane that fit the needs of target separation performances is a main issue. Firstly, various manufacturing processes to achieve best  $CO_2/CH_4$  separation performance are discussed, including important post-treatments. The formulation of polymer solution and effects of manufacturing parameters on the resulting membranes are presented. Other modifications, which involved PSf structure modification, PSf-based copolymer, mixed matrix membrane and facilitated transport membrane, are also discussed. Performances of those modified membranes are compared with the Robeson's upper bound and some modifications give satisfying results. Finally, commercial application of PSf membranes for  $CO_2/CH_4$  separation is discussed.

Keywords - CO<sub>2</sub> separation, polysulfone membrane

# 1. Introduction

One of the most widely investigated glassy polymer membrane materials for  $CO_2/CH_4$  separation is PSf. Polysulfone pure- and mixed-gas permeation properties have been extensively explored for gas separation due to its low price, chemical stability, and mechanical strength [1]. Compared to CA, PSf has lower CO<sub>2</sub> permeability and  $CO_2/CH_4$  selectivity [2] but higher plasticization pressure [3]. Plasticization pressure becomes important in practice due to its effect on membrane selectivity. Membrane with higher plasticization pressure can maintain its selectivity better than membrane with lower plasticization pressure on high CO<sub>2</sub> feed concentration or on high operation pressure.

Various types of PSf-based membrane for  $CO_2/CH_4$ separation have been developed but there are two limitations faced by the available membrane. Acquiring higher selectivity with at least similar productivity is the first challenge and the second challenge is maintaining membrane performance in the presence of complex and aggressive feed [4]. Many researches have been done to overcome those challenges by changing the membrane manufacturing process, formulation, and conduct material modification. In this work the variability of PSf-based membrane for  $CO_2/CH_4$  separation will be explored. The most important objective is to provide comprehensive information about PSf-based membrane for  $CO_2/CH_4$ separation.

### 2. Manufacturing processes

In general, there are three types of PSf membrane that can be used for  $CO_2/CH_4$  separation, dense, asymmetric, and composite. Dense and asymmetric membrane consists of PSf only while composite membrane consists of PSf and other polymer in different layer. Manufacturing processes of these types of membranes are different and versatile. For dense and asymmetric membrane there are three major process; dry, wet, and dry/wet. Dry process is so named because the membranes are formed without immersion in precipitation bath. In this method, the polymer solution consists of binary mixture between a polymer and a solvent or a mixture of polymers, solvents and nonsolvents. The polymer solution is then cast on a suitable support and the solvent is allowed to evaporate [2]. The PSf membrane obtained by dry process is a dense membrane. The gas permeation rate of this membrane is usually low due to the thickness of the selective layer [5].

In wet process, the polymer solution composition is the same with dry process. Prior to coagulation in a precipitation bath, a brief optional evaporation is often used to concentrate the outermost region of the membrane. The evaporation step does not induce the phase separation of the membrane. The phase separation phenomena occur in the coagulation bath due to the diffusional exchange of solvent and nonsolvent. The diffusional exchange causes the solution to be unstable thermodynamically and demixing occurs. In some cases, demixing occurs instantaneously and a porous membrane is obtained but in other cases, demixing occurs only after a certain period of time and produce membrane with asymmetric structure. Delay demixing can occur when nonsolvent with low mutual affinity is used. Even thought this process can produce PSf asymmetric membrane, generally the PSf membrane has too many defects to be used in gas separation without further treatment [5]. The most popular treatment is coating layer application. The coating material must have high flux so it will not give an additional resistance to gas transport, such as silicone rubber, the most popular coating material.

To produce PSf membrane with ultra thin defect free toplayer and negligible sublayer resistance, Pinnau and Koros proposed a new technique, dry/wet process. In this process, there are two solvents are used, a more volatile solvent and a less volatile solvent, which have different affinity to the nonsolvent. The cast film is then subjected to convective evaporation and free standing evaporation prior



to immersion in the precipitation bath. Convective evaporation step is believed to be the important step to produce defect free membrane. The schematic diagram of

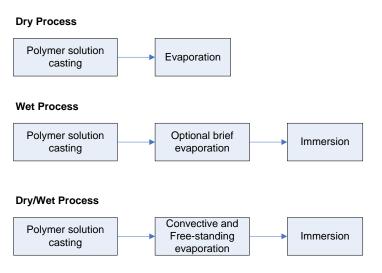


Figure 1 Schematic diagram of dry, wet, and dry/wet process

dry, wet, and dry/wet process is given in Fig.1. For processes that included immersion, selection of nonsolvent is important. Water is frequently used but when water is used, solvent exchange step is necessary. Solvent exchange is conducted to reduce the surface tension of water in membrane pores. High surface tension can cause pores wall collapse when water is removed and create a thick nonselective layer. Flat sheet composite membranes usually simply made by coating the second polymer onto the surface of the substrate polymer [6, 7, 8] and deposited the polymer for certain time while HF composite membranes made by dip-coating the substrate polymer [9, 10, 11] or co extrude spinning process [12, 13].

The asymmetric membrane manufacturing processes that have been discussed above are categorized as the standard procedures. To obtain defect free PSf membrane, it can also be made from dense PSf membranes by supercritical CO<sub>2</sub> process [14]. This process involves chloroform as a solvent. Chloroform was added to one side of the dense film then the film was immediately transferred into a high-pressure vessel and supercritical CO<sub>2</sub> was introduced and expanded. This process allows a very-controlled thickness of both the porous and dense layer. Factors affected those thickness are chloroform/Psf mass ratio,  $SCCO_2$ density, and depressurization rate of SCCO<sub>2</sub>. Due to difficulties to make defect free and food performance membrane, many studies about additional treatments in manufacturing processes have been conducted. Those treatments can be divided into two groups; first group is treatment during the manufacturing process of the membrane and the second group is treatments after the membrane is formed (post-treatments). To avoid defects formation during the manufacturing process of membranes, Shojaie et al. [15] proposed the usage of electric field. During the membrane manufacture, an optimum electric field, 2.2 kV/cm a.c., was maintained constant. The treated membranes show the average number density of defect decrease. In contrast, the total defect area is increase.

Gas separation membrane can be given several posttreatments to be modified physically or chemically and improve their separation characteristic [16]. Ilconich et al. [17] have studied the effect of ion beam irradiation on the structure and properties of asymmetric PSf membrane. The treated membranes possessed lower permeance and selectivity than those made conventionally. The energy used in this study was too high so it penetrates to the sublayer structure of the membrane so it caused a collapse of the porous sublayer structure and forms a thick non-selective resistance layer. Another study involve molecular bromine to form charge-transfer complex with unpair electrons along the PSf backbone. The molecular bromines in the treated membranes alter the free volume in such way so they achieve 100% selectivity increase with only 36% reduction in CO<sub>2</sub> permeability. The stability of those membranes was inconsistent so this treatment has not proved to be able to produce industrial membranes yet. Choi et al. [18] used ozone to oxidize and change the chemical structure of PSf membranes. The membranes were exposed to ozone containing oxygen for a certain time and those treated membranes show a significant increase of CO2/CH4 selectivity with CO<sub>2</sub> permeability decrease. Another study conducts surface fluorination of PSf asymmetric membranes. Those membranes only need to be exposed to the trace concentration of fluorine gas for several minutes to increase the CO<sub>2</sub>/CH<sub>4</sub> selectivity and decrease the CO<sub>2</sub> permeability [19, 20].

Hollow Fiber

Hollow Fiber



Forced convective residence time THF as a volatile									
Membrane type	P/I CO2 (GPU)	P/I CH4 (GPU)	Selectivity	Shear rate (/s)	FCRT <sup>a</sup> (s)	Coating	Reference		
Flat sheet	11.41	0.36	32.63	367	20	v	21		
Flat sheet	13.91	0.42	33				22		
Hollow Fiber	89	2.27	39.5	9710	0.71		23		

9710

8838

0.71

0.475

65.5

39.1

Table 1 Performance of PSf membranes made based on Pinnau and Koros formulation <sup>a</sup> Forced convective residence time THF as a volatile

 Table 2 Intrinsic permeation properties of dense PSf [22]

32.4

54

0.5

1.39

Permeability co [cm3(STP)cn	Intrinsic	
CO2	CH4	selectivity
4.5	0.16	28.1

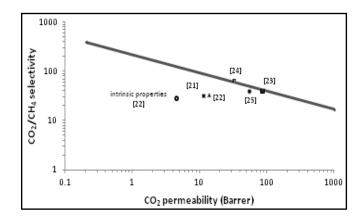
# 3. Membrane formulations and manufacturing parameters

Membrane formulation is a very crucial step to make a defect free and low resistance membrane for  $CO_2/CH_4$  separation. There are many aspects involve in the membrane formulation such as polymer concentration, influence of solvent ratio, effect of shear rate, influence of gelation media and conditions, effect of solvent evaporation condition, and additives. Many researches were done to find the optimum PSf membrane formulation. A formulation that has been used widely to make ultra thin defect free PSf membranes [21, 22, 23, 24, 25, 26] was given by Pesek and Koros [27]. Performances of those membranes are given in Table 1. The optimized polymer solution in dry/wet process consists of 22% PSf, 31.8% solvent, 31,8% DMAc as a less volatile solvent, and 14,4% ethanol as a nonsolvent.

The selectivity and permeability of the membranes in Table 1 are higher than the intrinsic selectivity of PSf as given in Table 2, so the defect free PSf membranes can be obtained by dry/wet process and Pinnau and Koros formulation. Beside defect free, this process and formulation, when combined with proper manufacturing parameters produce membrane which performance exceeded Robeson's upper bound as shown in Fig.2. To get a comprehensive understanding about how those parameters can affect the membrane performance, explanations are given below.

# 3.1 Polymer concentration

Optimum polymer concentration of polymer solution has to be determined in order to achieve defect free and thin selective layer for  $CO_2/CH_4$  separation process. Membrane casted from a dilute polymer solution produces a thin and porous selective layer that promotes high value of permeance but low selectivity. In contrast, higher polymer concentration in polymer solution leads to a denser and thicker selective layer that promotes higher selectivity but lower permeance [28, 29].



v

24

25

Figure 2 Performance of dry/wet process membrane

 Table 3 Water miscible solvent for hydrophobic asymmetric membrane [27]

Solvents	Boiling point (°C)
Acetone	56.5
Tetrahydrofuran	65.4
Acetonitrile	81.6
1,4-Dioxane	101.3
Dimethylformamide	153
Dimethylacetamide	165.2
Dimethylsulfoxide	189
N-Methylpyrrolidinone	202
γ-Butyrolactone	204.5
2-Pyrrolidinone	245

Interaction of solvent-polymer and nonsolvent-polymer can increase with increasing polymer concentration. Increase in solvent-polymer interaction reduces the coagulation value and increase in nonsolvent-polymer interaction reduces the dissolving power of the solvent. Those effects will promote delay demixing. In addition to those effects, increasing polymer concentration also increase the interaction of solvent and nonsolvent that retard diffusional exchange between solvent and nonsolvent, thus promote delay demixing too.

# 3.2 Influence of solvent ratio

In dry/wet membrane preparation process, a more volatile solvent introduced to the polymer solution to adjust the solvent evaporation and polymer coagulation well. Most common water miscible solvents used in hydrophobic asymmetric membrane are shown ini Table 3. Selection of a more volatile solvent is difficult due to some limitations.



More volatile solvent must have a good miscibility with water.

Whereas good water miscible solvent usually have a high degree of polarity and hydrogen bonding that suppress the solvent volatility. More volatile solvent is then removed during the evaporation process and in the outermost surface of the membrane generates a high polymer concentration. The more co-solvent evaporated, the thicker the concentrated polymer region that lead to the thicker selective layer of the membrane [28] and leads to a reduction in gas permeation rate. Furthermore, solvent ratio (ratio of more volatile solvent to less volatile solvent) can alter kinetics of phase inversion process depend on the degree of interaction between polymer and the solvents.

## **3.3** Effect of shear rate

Shear rate is defined as the ratio of velocity of the casting knife (or dope extrusion rate in the case of HF membrane) to membrane thickness. Increasing shear rate enhances molecular orientation in selective layer and offers a chance to increasing membrane selectivity [21, 24]. According to research by Sharpe et al. there is an optimum shear rate to produce defect free ultra thin PSf membrane. When the applied shear rate is too low, the membrane will have a thick selective layer while too high shear rate will produce thin and defective selective layer [23]. Literatures review on the effect of shear rate on membranes performance is shown in Table 4.

Membrane type	Shear rate (/s)	DER (cm3/min)	CO2 pressure normalized flux (cm3(STP)/s cm2 cmHg)	CO2/CH4 selectivity	Reference
	220		8.91	15.98	
PSF FS	275		11.7	24.99	21
	367		6	31.9	
PSf HF		1	17.6	39.1	24
		2.5	32.4	65.6	24
		0.8333	16	23.4	
		1.5	19.3	25.5	23
PSF HF		2.5	89	39.5	25
		3.33	61.9	12	
DCE EC	254		15.68	64.04	37
PSF FS	381		18.02	64.2	57

Table 4 Effect of shear rate on membranes performance

### 3.4 Influence of gelation medium and conditions

Gelation medium and conditions has a big influence on membrane structure and its gas transport properties. It is known that choice of gelation medium and condition can affect the speed of coagulation process. Mulder et al. were made PSf membrane by wet process and the results show that fast coagulation resulted in asymmetric membrane with thin porous selective layer while slow coagulation produced asymmetric membrane with dense and thick selective layer [30,31].

In order to produce an ultrathin defect free asymmetric FS membrane, Pinnau and Koros [32] studied the effect of different organic gelation media in dry/wet process. Methanol, ethanol, and 2-propanol were used as the gelation media. The study showed that methanol and ethanol quenched membrane had higher selectivity and flux than 2propanol quenched membrane. The results indicate that the phase separation and vitrification processes occurring during the wet process should be as rapid as possible. Another study was held by Yamasaki et al [29] using dual bath method. The first gelation media is 2-propanol and the second gelation media is water. The selective layer thickness determined by SEM is increase with the immersion time increase in the first bath thus changing the immersion time in the first bath can control the growth of the selective layer. Kim et al. [33] also studied the effect of dual bath gelation media on the morphology of the membrane. The first bath is polyethylene glycol (PEG) and the second bath is water. The results show that there is an optimum immersion time in PEG which lead to the formation of defect free selective layer. Radovanovic et al. [34] added solvent to the gelation

bath. As a result, the flux of nonsolvent across the phase interface is decreases with an increase of solvent concentration in the gelation bath because of the higher mass transfer resistance.

In the making of HF membrane, the influence of gelation media is more complicated than in the FS membrane because there are two different gelation media that can be used; the external and internal coagulant. Wang et al. [35] used water as the external coagulant. The internal coagulant used include water, ethanol, 2-propanol, the mixture of water/ethanol and water/2-propanol. The results show that membrane made with ethanol or 2-propanol as the internal coagulant show low permeability and selectivity. Ethanol and 2-propanol is known as weak nonsolvent for PSf so the internal coagulation rate and coagulant diffusion are very slow. This tends to increase the substructure resistance and thus reduce the permeability and selectivity.

# 3.5 Effect of solvent evaporation condition

In dry/wet process, convective evaporation is a crucial step to obtain ultrathin defect free asymmetric membrane. Some parameters in convective evaporation step, such as duration and temperature can influence the gas transport properties of a membrane. Typically, with the increase of convective evaporation duration or temperature, membrane skin thickness is also increase [36] and it leads to a lower  $CO_2$ permeance [25, 37]. According to Pesek and Koros [38], 10 to 15 s evaporation time are typically allowed to form ultrathin defect free membrane while only 0.5 to 1 s is needed for HF. In the making of hollow fiber membrane, the



duration of solvent evaporation can also be expressed as airgap length. The effect of evaporation duration in HF membrane is more complicated than FS (flat sheet) membrane. Nonsolvent from the lumen side penetrated deeper into the fiber and as the evaporation time increase, the nonsolvent can erode the skin layer of the membrane, causing thinner skin layer and a slight decrease in selectivity. Literatures review on the effect of solvent evaporation duration on membranes performance is shown in Table 5. Another research by Tsai et al. [39] shows that the air gap length and ambient humidity have dramatic effect on the membrane morphology. Macrovoids in the membrane can disappear, reappear and redisapper with increasing air-gap length.

3.6 Additives

The addition of additive into the polymer solution is known to changing solvent capacity [40, 41], phase inversion kinetics and thermodynamic properties [42]. The most widely used additive is polymers such as PEG and polyvinyl pyrolidone (PVP). Another additive can be nonsolvent, inorganic salts, mineral filler, etc. Zheng et al. studied effects of PEG introduction on kinetics of membrane formation [41]. From the experiment, it was found that PEG addition could enhance solution demixing thermodynamically and hindrance solution demixing rheologically. Another study gave characterization of PSf membrane with PEG as additive. It was seen that when molecular weight of PEG increases from 400 to 20000 Da, the mean pore size of the prepared membrane decreases, while the porosity and pore density increase [43].

Table 5	Effect of solvent	evaporation	duration on	membranes	performance
1 4010 0	Direct or sorreine	e aporation	addition on	memoranes	

Membrane type	FCRT (s)	Air gap length (cm)	CO2 pressure normalized flux (cm3(STP)/s cm2 cmHg)	CO2/CH4 selectivity	selective layer thickness (A)	Reference
	0.214	9	19.3			
PSf HF	0.357	15	12.6	30.6	726	25
	0.475	20	11.9	30.9	845	
	5		87.2	22.31	138	
	10		55.53	25.28	639	
PSF FS	15		6	31.9	929	
	20		11.41	32.63	800	21
	30		4.44	33.2	1724	21
	40		10.76	33.69	2286	
	50		6.71	34.54	2723	
	60		6.6	32.87	3595	
	0.237	5	82.2	36.8	848	
	0.426	9	58.2	40.1	1070	22
PSf HF	0.71	15	89	39.5	794	23
	0.947	20	84.4	26.3	590	
	15		15.68	64.04		37
PSF FS	18		12.17	67.33		37

Kang et al. studied the effect of nonsolvent addition on membrane morphology [44]. The nonsolvent additive was formic acid, water, or ethanol. The addition of nonsolvent to the polymer solution can bring the initial composition of the polymer solution nearer to the precipitation point. When a nonsolvent additive is added to polymer solution, it can decrease the dissolving power of the solvent and thus increase polymer-polymer interaction. By additive addition, the solvent concentration in polymer solution is decrease and the polymer chains have a smaller included volume because of a more tightly coiled conformation. Nonsolvent additive also influences the diffusion coefficient, which is a crucial parameter in controlling membrane morphology. When the diffusion coefficient of the nonsolvent is high for a given gelation media, the membrane consists of a smooth, defect free surface and macrovoid-free cross section regardless of the kind of the nonsolvent additive employed. The diffusion coefficient increases with an increase in nonsolvent additive concentration.

Comparison of the effect of PEG 400, PVP, ethanol, and glycerol was studied by Aroon et al. [45]. Additions of

additives bring the cloud point curves colder to the polymer-(solvent-additive) axis in the following order: PEG<ethanol<PVP<glycerol. The  $CO_2/CH_4$  separation factor improves in the same order of the position of the cloud point.

### **4** Structure related transport properties

The structure of PSf is affected the transport properties oh the membrane. The objective of researches effort on structure/property of PSf membranes is to create membranes with higher selectivity and permeability. Some researches try to fulfill this objective by change the structure of the commercial bisphenol A polysulfone and made polysulfone

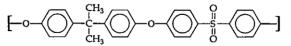


Figure 3 Primary chemical structure of bisphenol A PSf

copolymer meanwhile other researches blend the bisphenol A PSf and inorganic material to create mixed matrix material [4].



#### 4.1 Polysulfone structure modifications

The primary chemical structure of bisphenol A PSf is presented in Fig.3. Basically, all PSf modifications intend to change the free volume of the PSf chain and improve the gas transport properties. It was found that the CO<sub>2</sub> solubility coefficient and permeability increased with the increase in free volume [46]. Table 6 summarizes the results of those experiments. Effect of isopropylidene unit in bisphenol A PSf replacement was studied [48]. Aitken et al. [50] studied the effect of symmetry of phenylene linkages. In this study, the bisphenol A PSf was compared with the meta and para isomers of PSF, the meta and para isomers of TMPSF, and 3,4-PSf. The gas permeability coefficient trends for  $CO_2$  are in a good agreement with the FFV and shows that the meta isomers are the less permeable but more selective materials due to their unsymmetrical structure. Aitken et al. [51] also studied the comparison of PSf, TMPSF, 4,4'-bisphenol (BIPSF), and corresponding polysulfones with methyl ring substitutions, tetramethylbiphenol polysulfone (TMBIPSF), and hexamethylbiphenol polysulfone (HMBIPSF). The results show that BIPSF and PSF have very similar transport characteristics due to their similar packing behavior. Tetramethyl substitution increases permeability while hexamethyl substitution does not. Pixton and Paul [52] have been measured CO<sub>2</sub> and CH<sub>4</sub> transport properties in adamantane containing PSf. In all cases, the gas permeabilities and selectivity is higher in the adamantane containing PSf. Effects of basic substituents on gas transport properties of PSf have been studied [53] and the result shows that gas permeability in the aryl-substituted PSf were lower than the unmodified PSf. Ghosal et al. [54] studied the effect of aryl nitration on gas transport properties of PSf.The gas permeability on aryl nitration PSf is decreased but the selectivity shows a significant improvement. Zuniga et al. [55] created new PSf copolymers from bis(4fluorophenyl)sulfone and found that the newly-created

Table 6. Summary of permeability, selectivity, density, and FFV of the polymer modified membranes Permeability and selectivity measurement was done at 10 atm except for <sup>a</sup> 4 atm

Polymer	P CO <sub>2</sub> (Barrer)	CO <sub>2</sub> /CH <sub>4</sub> selectivity	Density	FFV	Reference
PSF	5.6	22	1.24	0.156	47
TMPSF	21	22	1.51	0.171	
DMPSF	2.1	30	1.213	0.149	
DMPSF-Z	1.4	34	1.227	0.136	
HFPSF	12	22	1.427	0.168	48
PSF-F	4.5	24	1.282	0.151	
PSF-O	4.3	24	1.33	0.15	
TMPSF-F	15	26	1.184	0.163	49
TMHFPSF	72	24	1.286	0.196	
3,4-PSF	1.5	29	1.25	0.149	50
PSF-P	6.8	20	1.191	0.156	
PSF-M	2.8	25	1.201	0.151	
TMPSF-P	13.2	22	1.127	0.168	
TMPSF-M	7	25	1.141	0.158	
BIPSF	5.6	22	1.291	0.154	51
TMBIPSF	31.8	25	1.195	0.164	
HMBISPSF	25.5	27	1.144	0.178	
1,3-ADM PSF	7.2	22	1.238	0.153	52
2,2-ADM PSF	9.5	24	1.231	0.158	
PSF-NH2 (16%)	2.7	24	1.253	0.134	53
PSF-NH2 (38%)	3.2	25	1.273	0.118	
PSF-CH2-NH2 (51%)	1.95	18	1.253	0.125	
PSF-CH2-imide (51%)	2.12	26	1.279	0.122	
PSF-NO2(50%)	3.4	24	1.272	0.135	54
PSF-NO2 (98%)	2.3	29	1.306	0.121	
PSF-NO2 (192%)	1.5	30	1.353	0.113	
TM-NPSF	4.85 <sup>°</sup>	36.7 <sup>a</sup>	1.231		55
HF-NPSF	4.89 <sup>a</sup>	33.5 <sup>°</sup>	1.364		
TMHF-NPSF	6.6 <sup>a</sup>	30.4 <sup>a</sup>			
1,5-NPSF	1.6	44	1.328		56
2,6-NPSF	1.5	41	1.328		
2,7-NPSF	1.8	36	1.339		
BPSF	3.2	27	1.514	0.156	57
MPSF	2.2	29	1.213	0.151	
TMSPSF	15.1	16	1.126	0.167	



polymers have superior gas transport properties compared to unmodified PSf.

Aitkin and Paul [56] studied the effect of naphthalene unit and its isomers on gas tansport properties of PSf. It is shown that the naphthalene unit increases the selectivity at the expense of permeability. The naphthalene unit position does not affect the gas transport properties significantly. The gas transport properties of brominated PSf membrane (BPSf) was examined and compared with trimethylsilylated bisphenol A PSf (TMSPSf), MPSf, and PSf [57]. The strong polarity of bromine reduces the chain packing-inhibiting ability. The permeability rank correlates well with FFV and the selectivity increases in the order. TMSPS<PSf<BPSf<MPSf. Unfortunately, PSf modifications do not give satisfying results. The permeability-selectivity combination values are still far under the Robeson's upper bound as shown in Fig.4.

### 4.2 Copolymer membranes

Another research to change the structure of PSf is creating PSf-based copolymer. The copolymer material may provide excellent properties of each polymer in one membrane. Poly

(ethylene oxide) is known as a polymer that has good affinity with  $CO_2$  molecules [58], thereby resulting in high solubility-selectivity. However, pure PEO tends to create crystalline region which is less permeable region for  $CO_2$ . PSf-PEO random copolymer was proposed to overcome this problem and produce membrane with high  $CO_2$  permeability as well as  $CO_2/CH_4$  selectivity. In fact, the permeability of  $CO_2$  in PSf-PEO membrane was reduced and the selectivity were not high due to the bad-phase structure separation of the copolymer.

### 4.3 Mixed matrix membranes

The mixed matrix is capable to combine the processability of polymers with the superior gas separation properties of rigid inorganic materials [59]. Polymer acts as the continuous phase or matrix and the inorganic material acts as the discrete phase. Schematic of a mixed matrix membrane is shown in Fig.5.

There are two kinds of inorganic material, porous and nonporous. Porous material inorganic enhance the gas separation properties by molecular sieving while non-porous inorganic material do that by conditioned the free volume of

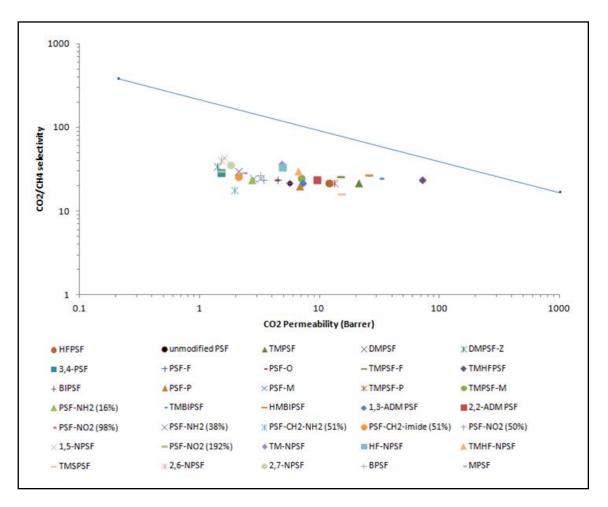


Figure 4 Performance of polysulfone-modified membranes



the matrix. The common inorganic materials used to produce MMM are zeolites [60, 61, 62, 63, 64], ordered mesoporous silica [65, 66], nonporous silica, carbon molecular sieves, and carbon nanotubes [67].

There are two major problems in the making of PSf MMM; the first one is inorganic material agglomeration and the second one is compatibility of the PSf and material inorganic. Both of those problems can reduce the selectivity of PSf MMM. To overcome the first problem, the composition of the inorganic material in PSf matrix should not be too much. Some action, such as inorganic molecular sizing [68], zeolite surface modification with silane [69], and particle functionalization [66, 70] can overcome the second problem.

Table 7 resumes the gas transport properties of MMM based on the type of inorganic material in the literature. The position of PSf-inorganic material permeability-selectivity combination values is presented in Fig.6 and only carbon black and silica-based inorganic material show interesting properties in terms of CO2 selectivity and/or permeability. In Wahab et al. experiment [72], the membrane structure made is asymmetric membrane; different with most of MMM experiment that made dense membrane structure as the continuous phase. Mass transfer resistance in asymmetric membranes is smaller than dense membrane so the CO<sub>2</sub> permeability can be higher. The optimum silica loading in this experiment is very low, 0.1%.wt. Particle agglomeration is unavoidable at higher silica loading and reduce the  $CO_2/CH_4$  selectivity.

# 4.4 Facilitated transport membranes

Facilitated transport membranes comprise a carrier with a special affinity toward a CO<sub>2</sub> gas molecule and this interaction controls the rate of transport. Wang et al. reported an excellent CO<sub>2</sub>/CH<sub>4</sub> selectivity through poly(Nvinyl-y-sodium aminobutyrate-co-sodium-acrylate) (VSA-SA)/PSf composite membrane [75]. VSA-SA has a secondary amine and a carboxylate ion, which can reversibly react with CO<sub>2</sub> and thus can be expected to act as carriers for CO<sub>2</sub>. In experiments with a mixed gas of 50 vol % CO<sub>2</sub> and 50 vol % CH<sub>4</sub>, the selectivity was 46.8 and the  $CO_2$  permeation rate was 9.2 x  $10^{-6}$  cm<sup>3</sup> (STP)/cm<sup>2</sup>s cmHg. Another research was held using Polyvinilamine (PVAm) as the selective layer [9]. The PVAm/PSf membrane was crosslinked by glutaraldehyde and sulfuric acid. As the result, the membrane posses higher CO<sub>2</sub>/CH<sub>4</sub> selectivity and lower CO<sub>2</sub> permeability. This membrane has an ideal

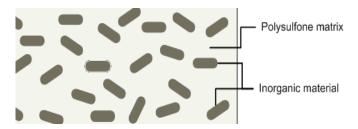


Figure 5 Schematic of a mixed matrix membrane

Inorganic material	wt-%	Permeabil	ity (barrer)	CO <sub>2</sub> /CH <sub>4</sub>	Reference
morganic materia	WL-70	CO2	CH4	selectivity	Reference
	0	1.3624	0.2833	4.809	
Zeolite ZSM-5	10	1.513	0.3419	4.4253	60
	20	1.5977	0.5505	2.9023	
	0	6.5	1.5	4.3	
Zeolite 13X	10	6.1	0.26	23.5	62
	20	6.1	0.32	19	
	0	3.9	0.17	23.55	
carbon nanotube	5	5.12	0.27	18.82	67
carbon nanotube	10	5.19	0.28	18.41	07
	15	4.52	0.28	16.09	
	0	4.5	0.17	23	70
Silica MCM-41	10	6.6	0.29	23	
	20	7.8	0.34	23	
	40	14.8	1	15	
	0	86.12	2.1	40.98	
carbon black <sup>a</sup>	2	76.25	2.16	35.4	71
	5	68.72	4.77	16.37	/1
	10	75.12	2.44	33.73	
	0	78.11	2.52	31.05	
fumed silica	0.1	90.04	2.75	32.74	72
rumeu sinca	3	88.06	3.57	24.64	12
	10	87.69	11.8	7.43	
	0	4.46	0.17	25.88	
Silica MCM-48	10	8.45	0.33	25.47	73
	20	18.21	0.77	23.58	

Table 7 MMM gas transport properties based on inorganic material type



fumed silica <sup>b</sup>	0	6.3	0.22	29	
	5	7.7	0.29	27	
	10	9.3	0.38	25	74
	15	12.9	0.62	21	
	20	19.7	1.1	18	

<sup>a</sup> Pressure normalized flux

<sup>b</sup> inorganic material in volume-%

separation factor of 206 and  $CO_2$  permeation rate of 3.99 x  $10^6$  cm<sup>3</sup>/cm<sup>2</sup>s cmHg. Wang et al. [76] studied gas transport property of polyallylamine (PAAm)-poly(vinyl alcohol) (PVA)/PSf composite membrane. PAAm contains amino group as the fixed carrier and this membrane shows performance that can exceed the Robeson's upper bound (Fig.6). For CO<sub>2</sub>/CH<sub>4</sub> separation, the selectivity reach 58 at 0.1 MPa feed gas pressure with 1.95 x  $10^{-5}$  cm<sup>3</sup>(STP)/cm<sup>2</sup>s KPa permeance.

# **5** Commercial application

The world market for natural gas is estimated at approximately US\$ 22 billion annually [4] and the market for membrane technology is getting bigger. Membrane technology can minimize the size of an amine system. Polysulfone membrane has been used commercially in  $CO_2/CH_4$  separation applications, in example Air Products PRISM membranes. Hollow fiber PRISM membranes are capable of withstanding differential pressure up to 1000 psi. Highest  $CO_2$  concentration in permeate, which is 95%, can be achieved from a feed containing 70%  $CO_2$ . The  $CO_2$ purity in permeate flow is decrease as the decrease of  $CO_2$  in the feed stream [77]. In July 2004, PRISM membrane technology was used in a  $CO_2$  removal system engineered and built by Petraco. Capacity of the plant is 8 MMSCFD and designed to reduce  $CO_2$  composition from 4.5% to less than 2%.

# 6. Conclusions

Abundant opportunities do indeed exist for PSf membrane for industrial  $CO_2/CH_4$  separation; however, performance improvement is still needed for the existing PSf membrane. Membrane formulation and manufacturing parameters have significant impact on membrane performance. Suitable formulation and manufacturing parameters even can produce membrane with performances that exceed Robeson's upper bound.

PSf modification also plays important role in order to achieve better membrane performance. Mixed matrix and facilitated transport membrane show a promising future in  $CO_2/CH_4$  separation despite of some technical barriers for industrial implementation. Overcome these barriers is a major goal to increase PSf membrane utilization in the field of  $CO_2/CH_4$  separation.

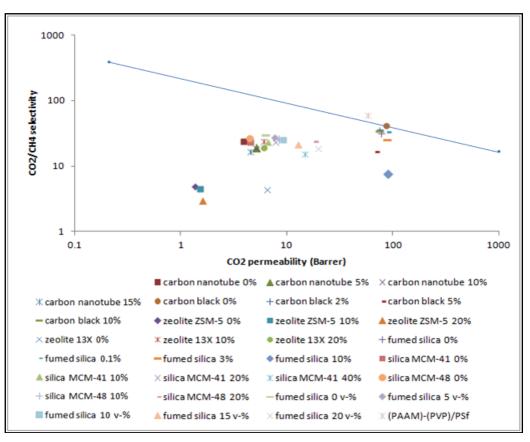


Figure 6 Performance of mixed matrix and facilitated transport membranes



#### References

- [1] H. Hachisuka, K. Ikeda, US Patent no 5 888 605, 1999.
- [2] M. H. V. Mulder, Basic principles of membrane technology (The Netherlands: Kluwer Academic Publishers, 1996).
- [3] A. Bos, I.G.M. Punt, M. Wessling, H. Strathmann, CO<sub>2</sub>-induced plasticization phenomena in glassy polymers, *Journal of Membrane Science*, 155, 1998, 57-78.
- [4] W.J. Koros, R. Mahajan, Pushing the limits on possibilities for large scale gas separation: which strategies?, *Journal of Membrane Science*, 175, 2000, 181-196.
- [5] I. Pinnau, W.J. Koros, Structures and gas separation properties of asymmetric membranes made by dry, wet, and dry/wet phase inversion, *Journal of Applied Polymer Science* 43, 1991, 1491-1502.
- [6] R. Du, X. Feng, A. Chakma, Poly (N,Ndimethylaminoethyl methacrylate)/polysulfone composite membranes for gas separations, Journal of Membrane Science 279, 2006, 76-85.
- [7] J. Marchese, N. Ochoa, C. Pagliero, Preparation and gas separation performance of silicone-coated polysulfone membranes, *J. Chem. Tech. Biotechnol.*, 63, 1995, 329-336.
- [8] R. Du, A. Chakma, X. Feng, Interfacially formed poly(N,N-dimethylaminoethylmethacrylate)/ polysulfone composite membranes for CO<sub>2</sub>/N<sub>2</sub> separation, *Journal of Membrane Science* 290, 2007, 19-28.
- [9] C. Dong, Z. Wang, C. Yi, S. Wang, Preparation of polyvinylamine/polysulfone composite hollow-fiber membranes and their CO<sub>2</sub>/CH<sub>4</sub> separation performance, *Journal of Applied Polymer Science 10*, 2006, 1885-1891.
- [10] Z. Ye, Y. Chen, H. Li, G. He, M, Deng, Preparation of a novel polysulfone/polyethylene oxide/silicone rubber multilayer composite membrane for hydrogen-nitrogen separation, *Material Chemistry and Physics 94*, 2005, 288-291.
- [11] L. Liu, A. Chakma, X. Feng, Preparation of hollow fiber poly(ether block amide)/polysulfone composite membranes for separation of carbon dioxide from nitrogen, *Chemical Engineering Journal 105*, 2004, 43-51.
- [12] X. Ding, Y. Cao, H. Zhao, L. Wang, Q. Yuan, Fabrication of high performance Matrimid/polysulfone dual-layer hollow fiber membranes for O<sub>2</sub>/N<sub>2</sub> separation, *Journal of Membrane Science 323*, 2008, 352-361.
- [13] S.S. Hosseini, N. Peng, T.S. Chung, Gas separation membranes developed through integration of polymer blending and dual-layer hollow fiber spinning process for hydrogen and natural gas enrichments, *Journal of Membrane Science* 349, 2010, 156-166.
- [14] A. Torres-Trueba, F.A. Ruiz-Trevino, G. Luna-Barcenas, C.H. Ortiz-Estrada, Formation of integrally skinned asymmetric polysulfone gas separation membranes by supercritical CO<sub>2</sub>, *Journal of Membrane Science 320*, 2008, 431-435.

- [15] S.S. Shojaie, A.R. Greenberg, W.B. Krantz, Use of an electric field to alter membrane morphology in a polysulfone-polyvinylpirolidone blend, *Journal of Membrane Science* 79, 1993, 115-122.
- [16] T.A. Barbari, S.S. Datwani, Gas separation properties of polysulfone membranes treated with molecular bromine, *Journal of Membrane Science* 107, 1995, 263-266.
- [17] J.B. Ilconich, X. Xu, M. Coleman, P.J. Simpson, Impact of ion beam irradiation on microstructure and gas permeance of polysulfone asymmetric membranes, *Journal of Membrane Science 214*, 2003, 143-156.
- [18] S-H. Choi, M-K. Lee, S-J. Oh, J-K. Koo, Gas Sorption and transport of ozone-treated polysulfone, *Journal of Membrane Science* 221, 2003, 37-46.
- [19] J.M. Mohr, D.R. Paul, I. Pinnau, W.J. Koros, Surface fluorination of polysulfone asymmetric membranes and films, *Journal of Membrane Science* 56, 1991, 77-98.
- [20] J.D. Le Roux, D.R. Paul, J. Kampa, R.J. Lagow, Modification of asymmetric polysulfone membranes by mild surface fluorination Part I. Transport properties, *Journal of Membrane Science* 94, 1994, 121-141.
- [21] A.F. Ismail, B.C. Ng, W.A.W. Abdul Rahman, Effects of shear rate and forced convection residence time on asymmetric polysulfone membranes structure and gas separation performance, *Separation and Purification Technology* 33, 2003, 255-272.
- [22] A.F. Ismail, W. Lorna, Suppression of plasticization in polysulfone membranes for gas separations by heattreatment technique, *Separation and Purification Technology 30*, 2003, 37-46.
- [23] I.D. Sharpe, A.F. Ismail, S.J. Shilton, A study of extrusion shear and forced convection residence time in the spinning of polysulfone hollow fiber membranes for gas separation, *Separation and Purification Technology* 17, 1999, 101-109.
- [24] A.F. Ismail, I.R. Dunkin, S.L. Gallivan, S.J. Shilton, Production of super selective polysulfone hollow fiber membranes for gas separation, *Polymer 40*, 1999, 6499-6506.
- [25] S.A. Gordeyev, S.J. Shilton, Forced convection spinning of gas separation hollow fibre membranes: some underlaying factors, mechanisms and effects, *Journal of Membrane Science 229*, 2004, 225-233.
- [26] S.A. Gordeyev, G.B. Lees, I.R. Dunkin, S.J. Shilton, Super-selective polysulfone hollow fiber membranes for gas separation: rheological assessment of the spinning solution, *Polymer 42*, 2001, 4347-4352.
- [27] S.C. Pesek, W.J. Koros, Aqueous quenched asymmetric polysulfone membranes prepared by dry/wet phase separation, *Journal of Membrane Science 81*, 1993, 71-88.
- [28] A.F. Ismail, P.Y. Lai, Effects of phase inversion and rheological factors on formation of defect-free and ultrathin-skinner asymmetric polysulfone membranes for gas separation, *Separation and Purification Technology 33*, 2003, 127-143.
- [29] A. Yamasaki, R.K. Tyagi, A.E. Fouda, T, Matsuura, K. Jonasson, Effect of gelation conditions on gas

**X** IOSR

separation performance for asymmetric polysulfone membranes, *Journal of Membrane Science 123*, 1997, 89-94.

- [30] M.H.V. Mulder, J. Oude Hendrikman, J. G. Wijmans, C.A. Smolders, A rationale for the preparation of asymmetric pervaporation membranes, *Journal of Applied Polymer Science* 30, 1985, 1539-1545
- [31] J.G. Wijmans, J. Kant, M.H.V. Mulder, C.A. Smolders, Phase separation phenomena in solutions of polysulfone in a mixture oa a solvent and a nonsolvent: relationship with membrane formation, *Polymer 26*, 1985, 1539-1545.
- [32] I. Pinnau, W.J. Koros, Influence of quench medium on the structures and gas permeation properties of polysulfone membranes made by wet and dry/wet phase inversion, *Journal of Membrane Science* 71, 1992, 81-96.
- [33] H.J. Kim, Tabe-Mohammadi, A. Kumar, A.E. Fouda, Asymmetric membranes by two stage gelation technique for gas separation: formation and characterization, Journal of Membrane Science 161, 1990, 229-238.
- [34] P. Radovanovic, S.W. Thiel, S-T. Hwang, Formation of asymmetric polysulfone membranes by immersion precipitation. Part I. Modelling mass transport during gelation, *Journal of Membrane Science* 65, 1992, 213-229.
- [35] D. Wang, W.K. Teo, K. Li, Preparation and characterization of high-flux polysulfone hollow fibre gas separation membranes, *Journal of Membrane Science 204*, 2002, 247-256.
- [36] A. Yamasaki, R.K. Tyagi, A.E. Fouda, T. Matsuura, K. Jonasson, Effect of solvent evaporation conditions on gas separation performance for asymmetric polysulfone membranes, *Journal of Applied Polymer Science 71*, 1999, 1367-1374.
- [37] A.F. Ismail, P.Y. Lai, Development of defect-free asymmetric polysulfone membranes for gas separation using response surface methodology, *Separation and Purification Technology* 40, 2004, 191-207.
- [38] S.C. Pesek, W.J. Koros, Aqueous quenched asymmetric polysulfone hollow fibers prepared by dry/wet phase separation, *Journal of Membrane Science* 88, 1994, 1-19.
- [39] H.A. Tsai, C.Y. Kuo, J.H. Lin, D.M. Wang, A. Deratani, C. Pochat-Bohatier, K.R. Lee, J.Y. Lai, Morphology control of polysulfone hollow fiber membranes via water vapor induced phase separation, *Journal of Membrane Science* 278, 2006, 390-400.
- [40] Q-Z. Zheng, P. Wang, Y-N. Yang, Rheological and thermodynamic variation in polysulfone solution by PEG introduction and its effect on kinetics of membrane formation via phase-inversion process, *Journal of Membrane Science* 279, 2006, 230-237.
- [41] S. Munori, A. Bottino, G. Capannelli, Preparation and characterization of polysulfone-polyvinilporrolidone based membrane, Desalination 70, 1988, 265-275.
- [42] P.S.T. Machado, A.C. Habert, C.L. Li, Membrane formulation mechanism based on precipitation kinetics and membrane morphology: flat and hollow fiber

polysulfone membranes, *Journal of Membrane Science* 155, 1999, 171-183.

- [43] B. Chakrabarty, A.K. Ghoshal, M.K. Purkait, SEM analysis and gas permeability test to characterize polysulfone membrane prepared with polyethylene glycol as additive, *Journal of Colloid and Interface Science 320*, 2008, 245-253.
- [44] Y.S. Kang, H.J. Kim, U.Y. Kim, Asymmetric membrane formation via immersion precipitation method. I. Kinetic effect, *Journal of Membrane Science* 60, 1991, 219-232.
- [45] M.A. Aroon, A.F. Ismail, M.M. Montazer-Rahmati, T. Matsuura, Morphology and permeation properties of polysulfone membranes for gas separation: Effects of non-solvent additives and co-solvent, *Separation and Purification Technology* 72, 2010, 194-202.
- [46] C-C. Hu, C-S. Chang, R-C. Ruaan, J-Y. Lai, Effect of free volume and sorption on membrane gas transport, *Journal of Membrane Science* 226, 2003, 51-61.
- [47] J.S. Mchattie, W.J. Koros, D.R. Paul, Gas transport properties of polysulphones: 1. Role of symmetry of methyl group placement on bisphenol rings, *Polymer* 32(5), 1991, 840-850.
- [48] J.S. Mchattie, W.J. Koros, D.R. Paul, Gas transport properties of polysulphones: 2. Effect of bisphenol connector groups, *Polymer 32(14)*, 1991, 2618-2625.
- [49] J.S. Mchattie, W.J. Koros, D.R. Paul, Gas transport properties of polysulphones: 3. Comparison of tetramethyl-substituted bisphenols, *Polymer 33(8)*, 1992, 1701-1711.
- [50] C.I. Aitken, W.J. Koros, D.R. Paul, Effect of structural symmetry on gas transport properties of polysulfones, *Macromolecules 25*, 1992, 3424-3434.
- [51] C.I. Aitken, W.J. Koros, D.R. Paul, Gas transport properties of bisphenol polysulfones, *Macromolecules* 25, 1992, 3651-3658.
- [52] M.R. Pixton, D.R. Paul, Gas transport properties of adamantane-based polysulfones, *Polymer 36(16)*, 1995, 3165-3172.
- [53] K. Ghosal, R.T. Chern, B.D. Freeman, W.H. Daly, I.I. Negulescu, Effect of basic substituents on gas sorption and permeation in polysulfone, *Macromolecules* 29, 1996, 4360-4369.
- [54] K. Ghosal, R.T. Chern, B.D. Freeman, R. Savariar, The effect of aryl nitration on gas sorption and permeation in polysulfone, *Journal of Polymer Science 33*, 1995, 657-666.
- [55] C. Camacho-Zuniga, F.A. Ruiz-Trevino, S. Hernández-López, M.G. Zolotukhin, F.H.J. Maurer, A. González-Montiel, Aromatic polysulfone copolymers for gas separation membrane applications, *Journal of Membrane Science 340*, 2009, 221-226.
- [56] C.L. Aitkin, D.R. Paul, Gas transport properties of polysulfones based on dihydroxynaphtalene isomers, *Journal of Polymer Science 31*, 1993, 1061-1065.
- [57] H-J. Kim, S-I. Hong, The transport properties of CO<sub>2</sub> and CH<sub>4</sub> for brominated polysulfone membrane, *Korean J. Chem. Eng.*, *16*(*3*), 1999, 343-350.
- [58] H.W. Kim, H.B. Park, Gas diffusivity, solubility, and permeability in polysulfone-poly(ethylene oxide)

IOSR

random copolymer membranes, *Journal of Membrane Science* 372, 2011, 116-124.

- [59] A. Brunetti, F. Scura, G. Barbieri, E. Drioli, Membrane technologies for CO<sub>2</sub> separation, *Journal of Membrane Science 35*, 2010, 115-125.
- [60] F. Dorosti, M.R. Omidkhah, M.Z. Pedram, F. Moghadam, Fabrication and characterization of polysulfone/polyimide-zeolite mixed matrix membrane for gas separation, *Chemical Engineering Journal 171*, 2011, 1469-1476.
- [61] P. Gorgojo, S. Uriel, C. Tellez, J. Coronas, Development of mixed matrix membranes based on zeolite Nu-6(2) for gas separation, *Microporous and Mesoporous Materials* 115, 2008, 85-92.
- [62] T.M. Gur, Permselectivity of zeolite filled polysulfone gasseparation membranes, *Journal of Membrane Science 93*, 1994, 283-289.
- [63] B. Zornoza, O. Esekhile, W.J. Koros, C. Tellez, J. Coronas, Hollow silicate-1 sphere-polymer mixed matrix membranes for gas separation, *Separation and Purification Technology* 77, 2011, 137-145.
- [64] L.Y. Jiang, T.S. Chung, S. Kulprathipanja, An investigation to revitalize the separation performance of hollow fibers with a thin mixed matrix composite skin for gas separation, *Journal of Membrane Science* 276, 2006, 113-125.
- [65] B. Zornoza, S. Irusta, C. Tellez, J. Coronas, Mesoporous silica sphere-polysulfone mixed matrix membranes for gas separation, *Langmuir 25(10)*, 2009, 5903-5909.
- [66] A. Jomekian, M. Pakizeh, A.R. Shafiee, S.A.A. Mansoori, Fabrication or preparation and characterization of new modified MCM-41/PSf nanocomposite membrane coated by PDMS, *Separation and Purification Technology* 80, 2011, 556-565.
- [67] S. Kim, L. Chen, J.K. Johnson, E. Marand, Polysulfone and functionalized carbon nanotube mixed matrix membranes for gas separation: Theory and experiment, *Journal of Membrane Science 294*, 2007, 147-158.
- [68] W.A.W. Rafizah, A.F. Ismail, Effect of carbon molecular sieve sizing with poly(vinyl pyrrolidone) K-

15 on carbon molecular sieve-polysulfone mixed matrix membrane, *Journal of Membrane Science 307*, 2008, 53-61.

- [69] A.L. Khan, A. Cano-Odena, B. Gutierrez, C. Minguillon, I.F.J. Vankelecom, Hydrogen separation and purification using polysulfone acrylate-zeolite mixed matrix membrane, *Journal of Membrane Science 350*, 2010, 340-346.
- [70] S. Kim, E. Marand, High permeability nano-composite membranes based on mesoporous MCM-41 nanoperticles in a polysulfone matrix, *Microporous and Mesoporous Materials* 114, 2008, 129-136.
- [71] V. Bhardwaj, A. Machintosh, I.D. Sharpe, S.A. Gordeyev, S.J. Shilton, Polysulfone hollow fiber gas separation membranes filled with submicron particles, *Ann. N. Y. Acad. Sci.* 984, 2003, 318-328.
- [72] M.F.A. Wahab, A.F. Ismail, S.J. Shilton, Studies on gas permeation performance of asymmetric polysulfone hollow fiber mixed matrix membranes using nanosized fumed silica as fillers, *Separation and Purification Technology xxx*, 2011, xxx-xxx.
- [73] S. Kim, E. Marand, Polysulfone and mesoporous molecular sieve MCM-48 mixed matrix membranes for gas separation, *Chem. Mater.*, 18, 2006, 1149-1155.
- [74] J. Ahn, W-J. Chung, I. Pinnau, M.D. Guiver, Polysulfone/silica nanoparticle mixed-matrix membranes for gas separation, *Journal of Membrane Science 314*, 2008, 123-133.
- [75] Z. Wang, C. Yi, Y. Zhang, J. Wang, S. Wang, CO<sub>2</sub>facilitated transport through poly(N-vinyl-γ-sodium aminobutyrate-co-sodium acrylate)/polysulfone composite membranes, *Journal of Applied Polymer Science 100*, 2006, 275-282.
- [76] Y. Cai, Z. Wang, C. Yi, Y. Bai, J. Wang, S. Wang, Gas transport property of polyallylamine-poly(vinyl alcohol)/polysulfone composite membranes, *Journal of Membrane Science* 310, 2008, 184-196.
- [77] D.J. Stookey, T.E. Graham, W.M. Pope, Natural gas processing with PRISM® separators, *Environmental Progress 3*, 1984, 212-214.